# $\mathrm{R}_{12} \mathrm{Co}_{7}$ Compounds with $\mathrm{R}=\mathrm{Gd}$, Tb, Dy, $\mathrm{Ho}, \mathrm{Er}$ 

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#### Abstract

Ho}_{12} \mathrm{Co}_{7}\) crystallizes with a new monoclinic structure type. Space group $P 2_{1} / c$ (No. 14); $a=8.327$ (6), $b=11 \cdot 191$ (4), $c=13.871$ (5) $\AA, \beta=138 \cdot 8$ (1) ${ }^{\circ} ; Z=2$, $D_{m}=9 \cdot 51, D_{x}=9 \cdot 33 \mathrm{~g} \mathrm{~cm}^{-3} ; \mu$ (Мо $K \alpha$ ) $=485 \mathrm{~cm}^{-1}$. Weissenberg films, Patterson method, absorption correction, least-squares refinement. $R=0.08$ for 850 in dependent reflections. $\mathrm{Gd}_{12} \mathrm{Co}_{7}, \mathrm{~Tb}_{12} \mathrm{Co}_{7}, \mathrm{Dy}_{12} \mathrm{Co}_{7}$ and $\mathrm{Er}_{12} \mathrm{Co}_{7}$ are isostructural with $\mathrm{Ho}_{12} \mathrm{Co}_{7}$. The four different types of Co-centered rare-earth polyhedra are identical to the four different types of Rh -centered rare-earth polyhedra found in $\mathrm{Y}_{3} \mathrm{Rh}_{2}$. They are a trigonal prism, a cube, an Archimedian antiprism and a truncated Archimedian antiprism.


Introduction. Phase diagrams for the systems $\mathrm{Gd}-\mathrm{Co}$, Dy-Co, $\mathrm{Ho}-\mathrm{Co}, \mathrm{Er}-\mathrm{Co}$ and Y-Co have been reported by Buschow (1971). At $37 \mathrm{at} . \%$ Co all these diagrams show the presence of a phase labeled $\mathrm{R}_{x} \mathrm{Co}(\mathrm{R}=\mathrm{Gd}$, Dy, Ho, Er, Y) with congruent melting behavior. In fact three different compounds have been identified in the Y-Co system with compositions close to $40 \mathrm{at} . \%$ Co, namely hexagonal $\mathrm{Y}_{4} \mathrm{Co}_{3}$ (Lemaire, Schweizer \& Yakinthos, 1969), orthorhombic $\mathrm{Y}_{3} \mathrm{Co}_{2}$ (Moreau, Parthé \& Paccard, 1975) and monoclinic $\mathrm{Y}_{8} \mathrm{Co}_{5}$ (Moreau, Paccard \& Parthé, 1976a). In the systems with Gd, Tb, Dy, Ho and Er only two compounds are found, one being isotypic with hexagonal $\mathrm{Y}_{4} \mathrm{Co}_{3}$ and the other crystallizing with an unknown crystal structure. The purpose of this paper is to present the results of a structure determination of this phase, for which the exact composition has been found to be $\mathrm{R}_{12} \mathrm{Co}_{7}$ with $36 \cdot 8 \mathrm{at} . \% \mathrm{Co}$.
The alloys were made from commercially available elements of high purity: rare earth $99.9 \%$ and Co

[^0]$99.99 \%$. Samples of $\mathrm{Ho}_{12} \mathrm{Co}_{7}$ were prepared by conventional arc-melting techniques and then annealed for 7 days at $750^{\circ} \mathrm{C}$. From the crushed button, a single crystal was isolated, with a pseudo-cylindrical shape: a long axis of 0.157 mm and a diameter varying between 0.067 and 0.095 mm . The diffraction study was carried out on this crystal.

Weissenberg photographs showed the crystal to be monoclinic with the unequivocal space group $P 2_{1} / c$ (systematic absences, $h 0 l$ with $l$ odd and $0 k 0$ with $k$ odd).

Lattice constants and intensities were measured on Weissenberg films with Zr -filtered Mo $K \alpha$ radiation. Intensities were estimated by comparison with a standard. A cylindrical absorption correction was made with a calculated value of $\mu R=2 \cdot 3$, corresponding to an average diameter of 0.081 mm . An empirical correction based on the comparison of several equivalent reflections was also applied to take into account the varying diameter of the cylinder.

The volume of the unit cell together with density measurements and space-group restrictions indicate that there are two formula units of $\mathrm{Ho}_{12} \mathrm{Co}_{7}$ per cell. Consequently it might be expected that the Ho atoms are in six different crystallographic sites with a multiplicity of 4 (general position $4 e$ ), and the Co atoms in three general sites with multiplicity 4 and one special site with multiplicity 2.

The measured intensities were used to compute a sharpened Patterson function. Examination of Harker sections allowed the positioning of the six Ho atoms. From an electron density map computed with the observed amplitudes and the phases due to the contribution of Ho atoms it was possible to locate all expected Co atoms. All positional and isotropic thermal parameters refined satisfactorily by least squares. The scattering factors of International Tables for X-ray Crystal-
lography (1968) were used. The final $R\left(=\Sigma|\Delta F| / \Sigma\left|F_{\mathrm{o}}\right|\right)$ calculated with 850 observed reflections was 0.081 . The final atomic parameters are listed in Table 1. Interatomic distances around all Co atoms are given in Table 2. All distances are similar to those found in rare-earth-Co compounds, except one very short $\mathrm{Co}(1)-\mathrm{Co}(3)$ distance of $2 \cdot 32 \AA$. However, a slightly larger Co -Co distance of $2.35 \AA$ has been found in $\mathrm{Y}_{8} \mathrm{Co}_{5}$ (Moreau, Paccard \& Parthé, 1976a) and of $2.37 \AA$ also in $\mathrm{Y}_{8} \mathrm{Co}_{5}$ and in $\mathrm{Y}_{3} \mathrm{Co}_{2}$ (Moreau, Parthé \& Paccard, 1975). As this structure is of a new type a listing of the low-angle reflections with corresponding intensities for X-ray powder diagram identification is given in Table 3 (Yvon, Jeitschko \& Parthé, 1969).*

Table 1. Atomic parameters for $\mathrm{Ho}_{12} \mathrm{Co}_{7}$ with e.s.d.'s in parentheses
The Debye-Waller factor is defined as $\exp \left[-B(\sin \theta / \lambda)^{2}\right]$. Space group $P 2_{1} / c$.

|  | Equi- <br> point | $x$ | $y$ | $z$ | $B(\AA)^{2}$ |
| :--- | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Ho}(1)$ | $4 e$ | $0.8799(7)$ | $0.5704(3)$ | $0.8031(4)$ | $1.50(7)$ |
| $\mathrm{Ho}(2)$ | $4 e$ | $0.5982(7)$ | $0.2027(3)$ | $0.1729(4)$ | $1.44(7)$ |
| $\mathrm{Ho}(3)$ | $4 e$ | $0.9449(7)$ | $0.5719(3)$ | $0.5863(4)$ | $1.54(7)$ |
| $\mathrm{Ho}(4)$ | $4 e$ | $0.6879(7)$ | $0.2957(3)$ | $0.4604(4)$ | $1.63(7)$ |
| $\mathrm{Ho}(5)$ | $4 e$ | $0.8213(7)$ | $0.8400(3)$ | $0.6603(4)$ | $1.50(7)$ |
| $\mathrm{Ho}(6)$ | $4 e$ | $0.5565(7)$ | $0.0050(3)$ | $0.3472(4)$ | $1.59(7)$ |
| $\mathrm{Co}(1)$ | $4 e$ | $0.787(2)$ | $0.694(1)$ | $0.348(1)$ | $2.8(2)$ |
| $\mathrm{Co}(2)$ | $4 e$ | $0.678(2)$ | $0.089(1)$ | $0.594(1)$ | $1.6(2)$ |
| $\mathrm{Co}(3)$ | $4 e$ | $0.976(2)$ | $0.836(1)$ | $0.529(1)$ | $3.1(3)$ |
| $\mathrm{Co}(4)$ | $2 b$ | 2 | 0 | 0 | $2.3(3)$ |

The lattice parameters of $\mathrm{Gd}_{12} \mathrm{Co}_{7}, \mathrm{~Tb}_{12} \mathrm{Co}_{7}, \mathrm{Dy}_{12} \mathrm{Co}_{7}$, $\mathrm{Ho}_{12} \mathrm{Co}_{7}$ and $\mathrm{Er}_{12} \mathrm{Co}_{7}$ are reported in Table 4. These parameters were obtained with PARAM (X-RAY system, 1972). The reflections were measured on films calibrated with Si , taken with a Guinier-de Wolff camera and $\mathrm{Cu} K \alpha$ radiation. The variation of the cell parameters is a consequence of the normal lanthanide contraction. Patterns were difficult to obtain from powder specimens. Preparation of powders by crushing apparently destroyed the crystalline nature of the samples and powder patterns could only be obtained after heat treatment of the crushed melt. The resulting patterns lacked resolution but were adequate for identification of the structure involved.

Discussion. A projection of the $\mathrm{Ho}_{12} \mathrm{Co}_{7}$ structure on the (100) plane is shown in Fig. 1. One recognizes three types of Co-centered rare-earth polyhedra. $\mathrm{Co}(1)$ is at the center of a trigonal prism, $\mathrm{Co}(2)$ is surrounded by an Archimedian antiprism and $\mathrm{Co}(4)$ is at the center of a cube. For clarity of presentation the coordination figure around $\mathrm{Co}(3)$ has not been indicated. If $3 \cdot 10 \AA$

[^1]Table 2. Interatomic distances of Co atoms in $\mathrm{Ho}_{12} \mathrm{Co}_{7}$ up to $4 \AA$
All e.s.d.'s are less than $0.02 \AA$.

| $\mathrm{Co}(1)-\mathrm{Co}(3)$ | 2.32 | $\mathrm{Co}(3)-\mathrm{Co}(1)$ | 2.32 |
| :---: | :---: | :---: | :---: |
| * $\mathrm{Ho}(3)$ | 2.82 | $\dagger \mathrm{Ho}(1)$ | 2.79 |
| * $\mathrm{Ho}(5)$ | 2.85 | $\dagger \mathrm{Ho}(2)$ | 2.79 |
| * Ho(6) | 2.89 | $\dagger \mathrm{Ho}(5)$ | 2.92 |
| * $\mathrm{Ho}(1)$ | 2.94 | $\dagger \mathrm{Ho}(6)$ | 2.98 |
| * $\mathrm{Ho}(4)$ | 2.95 | $\dagger \mathrm{Ho}(4)$ | 3.07 |
| * $\mathrm{Ho}(2)$ | 2.99 | $\dagger \mathrm{Ho}(1)$ | 3.09 |
| $\mathrm{Ho}(4)$ | $3 \cdot 13$ | $\mathrm{Ho}(3)$ | $3 \cdot 12$ |
| $\mathrm{Ho}(3)$ | $3 \cdot 41$ | $\mathrm{Ho}(6)$ | $3 \cdot 35$ |
| $\mathrm{Ho}(6)$ | $3 \cdot 97$ | $\mathrm{Ho}(2)$ | $3 \cdot 57$ |
|  |  | $\mathrm{Co}(3)$ | 3.84 |
| $\mathrm{Co}(2)-\ddagger \mathrm{Ho}(1)$ | 2.78 | $\mathrm{Co}(4)-28 \mathrm{Ho}(1)$ | 2.88 |
| $\mathrm{Co}(2)$ | 2.81 | $2 \mathrm{SHO}(2)$ | 2.93 |
| $\ddagger$ Но(6) | $2 \cdot 82$ | $28 \mathrm{Ho}(3)$ | 3.02 |
| $\ddagger \mathrm{Ho}(2)$ | $2 \cdot 86$ | $28 \mathrm{Ho}(4)$ | 3.05 |
| $\ddagger \mathrm{Ho}$ (5) | $2 \cdot 88$ |  |  |
| $\ddagger \mathrm{Ho}$ (5) | $2 \cdot 89$ |  |  |
| $\ddagger \mathrm{Ho}$ (6) | $2 \cdot 90$ |  |  |
| $\ddagger \mathrm{Ho}$ (3) | $2 \cdot 93$ |  |  |
| $\ddagger \mathrm{Ho}$ (4) | 3.00 |  |  |
| Ho(5) | $3 \cdot 94$ |  |  |
| Trigonal prism. |  |  |  |
| Pentagon with one corner replaced by a pair of atoms |  |  |  |
| endicular to the pentagon plane. Archimedian antiprism. |  |  | Archimedian antiprism. |
| Cube. |  |  |  |

Table 4. Lattice constants for $\mathrm{R}_{12} \mathrm{Co}_{7}$ compounds with space group $P 2_{1} / c$
All e.s.d.'s are less than $0.05 \AA$. $V=$ volume of the unit cell. $n=$ number of atoms in the unit cell.


Fig. 1. A projection of the $\mathrm{Ho}_{12} \mathrm{Co}_{7}$ structure on the (100) plane showing three of the four different types of Cocentered rare-earth polyhedra.
is to be considered the upper limit of the acceptable coordination distances, $\mathrm{Co}(3)$ is surrounded by six Ho atoms. The coordination can be described by two equivalent figures: either a truncated Archimedian antiprism with two opposite corners through the center of the antiprism missing or a pentagon with one corner being replaced by a pair of atoms along a direction perpendicular to the pentagon plane. The occurrence of these different coordination polyhedra with $\mathrm{Ho}_{12} \mathrm{Co}_{7}$ is rather unexpected, since the compounds on both sides, $\mathrm{Ho}_{3} \mathrm{Co}$ and $\mathrm{Ho}_{4} \mathrm{Co}_{3}$, have a much simpler construction. They are all composed of Co -centered trigonal rare-earth prisms. However, in $\mathrm{Ho}_{4} \mathrm{Co}_{3}$ one of the nine Co atoms per unit cell is at the center of an octahedron. The structure of $\mathrm{Y}_{3} \mathrm{Rh}_{2}$ (Moreau, Paccard \& Parthé, 1976b) is characterized by the same four types of transition metal-centered rare-earth polyhedra found in $\mathrm{Ho}_{1_{2}} \mathrm{Co}_{7}$.

The $\mathrm{R}_{3} \mathrm{Rh}_{2}$ alloys are brittle whilst those of the $\mathrm{R}_{12} \mathrm{Co}_{7}$ phase show typical plastic behavior. Hence there seems to be no correlation between the mechanical properties and the type of coordination polyhedra of the transition metal atoms.

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## References

[^2]
# N -2-(6'-Methyl)pyridyl-4-ethoxycarbonyl-2 $\mathrm{H}, \mathbf{3 H}$-pyrrole-2-one 

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#### Abstract

C}_{13} \mathrm{H}_{14} \mathrm{~N}_{2} \mathrm{O}_{3}\), monoclinic, $P 2_{1} / c \quad C_{2 h}^{5}, a=$ 8.840 (2), $b=7.880$ (2), $c=17.814$ (4) $\AA, \beta=97.21$ (1) ${ }^{\circ}$, $D_{x}=1.32$ (1), $D_{c}=1.328 \mathrm{~g} \mathrm{~cm}^{-3}, \quad Z=4$. The compound was prepared and kindly supplied by Vasvári-Debreczy and Mészáros at the Chinoin Research Laboratories, Budapest. Recrystallization from $\mathrm{CHCl}_{3}$ yielded suitable crystals. The final $R$ index is 0.048 for the full-matrix least-squares refinement of 164 variables based on 2074 observations. The dihedral angle between the best planes of the five and sixmembered rings is $5 \cdot 2^{\circ}$.

Introduction. As part of a systematic study of pharmaceutically interesting $N$-(pyridyl)pyrrole derivatives Dvortsák, Náray-Szabó \& Vasvári-Debreczy (1975) have correlated PMR chemical shift data with quantum mechanical calculations of charge distributions. The present crystal structure study was undertaken to

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provide details on the conformation in the solid state of one such derivative.
Details of data collection on a Picker FACS-I diffractometer are given in Table 1 and parallel those described earlier (Corfield, Doedens \& Ibers, 1967). The unit-cell parameters were determined from a leastsquares refinement of 12 reflections in the range $50<$ $2 \theta\left(\mathrm{Cu} K \alpha_{1}\right)<58^{\circ}$. No decomposition occurred during data collection. The data were processed in the usual way, using a value of $p$ of 0.04 in the estimation of standard deviations. The structure was solved by direct methods and refined by full-matrix least-squares methods utilizing the 2074 reflections above background. In the refinement the quantity $\sum w\left(\left|F_{o}\right|-\left|F_{c}\right|\right)^{2}$ was minimized, where $w=4 F_{o}^{2} / \sigma^{2}\left(F_{o}^{2}\right)$. Scattering factors for the nonhydrogen atoms were taken from Cromer \& Waber (1974) while those for H were from Stewart, Davidson \& Simpson (1965). From the initial $E$ map the positions of all nonhydrogen atoms were clear. On a subsequent difference Fourier map the positions of all the H atoms were discernible. These


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[^1]:    * A list of structure factors and Table 3 have been deposited with the British Library Lending Division as Supplementary Publication No. SUP 31807 ( $13 \mathrm{pp} ., 1$ microfiche). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 13 White Friars, Chester CH1 1NZ, England.

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